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## Platinum Bronzes. II.<sup>1</sup> Crystal Structures of $\text{CaPt}_2\text{O}_4$ and $\text{Cd}_{0.3}\text{Pt}_3\text{O}_4$

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The crystal structures of  $\text{CaPt}_2\text{O}_4$  and  $\text{Cd}_{0.3}\text{Pt}_3\text{O}_4$  have been determined from X-ray and neutron diffraction powder data. Refinement of both X-ray and neutron intensity data led to final weighted and unweighted agreement indices of 24.6 and 16.0% for  $\text{CaPt}_2\text{O}_4$  and 6.8 and 9.8% for the simpler  $\text{Cd}_{0.3}\text{Pt}_3\text{O}_4$ .  $\text{CaPt}_2\text{O}_4$  has a structure that is the two-dimensional analog of the general platinum bronze structure. It crystallizes in space group  $D_{4h}^9\text{-}P4_2/mmc$  of the tetragonal system with two formula units in a cell of dimensions  $a = 5.7786$  (5) Å and  $c = 5.5989$  (7) Å. The structure appears to be a unique one involving nonintersecting Pt chains in the  $x$  and  $y$  directions. Pt chain compounds in one and three dimensions are known and  $\text{CaPt}_2\text{O}_4$  forms the missing two-dimensional link.  $\text{Cd}_{0.3}\text{Pt}_3\text{O}_4$  possesses the  $\text{NiPt}_3\text{O}_4$  structure involving Pt chains in the  $x$ ,  $y$ , and  $z$  directions. It crystallizes in space group  $O_h^3\text{-}Pm\bar{3}n$  of the cubic system with two formula units in a cell with a cell edge of 5.6423 (4) Å. The unusual formal oxidation states of Pt in both compounds are explained by extensive metallic bonding between the Pt atoms.

### Introduction

Interest in the oxides of platinum arises first of all from their very existence. Platinum metal is used extensively as an inert material in high-temperature chemistry. Knowledge of the conditions under which the metal will react chemically is therefore of considerable importance. A further incentive for study of these oxides is the catalytic activity of some ill-defined platinum oxides, as exemplified by Adams' catalyst.<sup>3</sup>

Platinum bronzes are ternary oxides of platinum with the general formula  $\text{M}_x\text{Pt}_3\text{O}_4$ , where M is generally a group I or II cation and  $x$  ranges from zero to one. Palladium also forms this type of oxide. With Pd as the central metal, the Na, Ca, Cd, and Sr bronzes have been synthesized, while with Pt, the Na, Mg, and Ni bronzes are known.<sup>1</sup> While attempting to prepare  $\text{CaPt}_3\text{O}_4$  we obtained the phases  $\text{CaPt}_2\text{O}_4$  and  $\text{Ca}_4\text{PtO}_6$ . Both these materials were first prepared by Post<sup>4</sup> and later by McDaniel.<sup>5</sup> Czaya accidentally prepared single crystals of  $\text{Ca}_4\text{PtO}_6$  during syntheses of calcium silicates in a platinum crucible using a  $\text{CaCl}_2$  flux.<sup>6</sup> All of our attempts to prepare single crystals of  $\text{CaPt}_2\text{O}_4$  at atmospheric pressure have been fruitless.  $\text{CdPt}_3\text{O}_4$  has not been reported before. We have prepared both  $\text{Cd}_{0.3}\text{Pt}_3\text{O}_4$  and  $\text{Cd}_1\text{Pt}_3\text{O}_4$ , thereby proving for the first time that  $x$  in  $\text{M}_x\text{Pt}_3\text{O}_4$  is truly a variable, as was first suggested by Waser and McClanahan for  $\text{Na}_x\text{Pt}_3\text{O}_4$ .<sup>7</sup> No single crystals were obtained.

The unit cell of  $\text{CaPt}_2\text{O}_4$  suggested that its structure might be related to that of  $\text{Ni}_x\text{Pt}_3\text{O}_4$ ,<sup>1</sup> *i.e.*, the general platinum bronze structure. From initial inspection of the powder diffraction pattern it was clear that only a few variables would have to be determined in order to solve the structure. The pattern furthermore showed no resemblance to that of any of the known  $\text{AB}_2\text{O}_4$  structures. Therefore, a combined X-ray and neutron powder diffraction study was undertaken.

Only  $\text{Cd}_{0.3}\text{Pt}_3\text{O}_4$  could be prepared in sufficient quantity to carry out a neutron diffraction study. The X-ray powder diffraction pattern of  $\text{Cd}_{0.3}\text{Pt}_3\text{O}_4$  shows that it has the general platinum bronze structure. Since Cd is a better X-ray

scatterer than Ni or Na, we hoped to find out in the present instance whether the counterions ( $\text{Cd}^{2+}$ ) are ordered or randomly distributed over the available lattice sites. Furthermore, because the earlier X-ray diffraction study on  $\text{Ni}_x\text{Pt}_3\text{O}_4$  failed to distinguish between a centric and noncentric structure, an X-ray and neutron study was carried out on  $\text{Cd}_{0.3}\text{Pt}_3\text{O}_4$  in order to try to resolve these problems.

### Experimental Section

Both  $\text{CaPt}_2\text{O}_4$  and  $\text{Cd}_{0.3}\text{Pt}_3\text{O}_4$  were prepared by the reaction of  $(\text{NH}_4)_2\text{PtCl}_6$  with the corresponding carbonate under an atmosphere of  $\text{O}_2$ .<sup>8</sup>

X-Ray powder diffraction data for  $\text{CaPt}_2\text{O}_4$  were collected in the continuous scan mode on a Picker diffractometer. Monochromatic  $\text{Co K}\bar{\alpha}$  radiation was obtained from the (200) plane of a LiF crystal. X-Ray powder data for  $\text{Cd}_{0.3}\text{Pt}_3\text{O}_4$  were collected in the step-scan mode on a Norelco diffractometer. Monochromatic  $\text{Cu K}\bar{\alpha}$  radiation was obtained from the (002) plane of a graphite crystal. In both instances the samples were approximately 2 mm thick and were in the flat-plate geometry. Accordingly, no correction for absorption is necessary.<sup>9</sup>

The neutron diffraction data were obtained with the Argonne computer aided diffraction equipment<sup>10</sup> at the CP-5 reactor. The data for  $\text{CaPt}_2\text{O}_4$  were collected on the heavy-duty powder unit with  $\lambda = 1.20$  Å obtained from the (111) plane of a germanium monochromator. Two  $\theta$  step scans ( $0.2^\circ$  step) were timed with a fission monitor (approximately 15 min/step). Data for  $\text{Cd}_{0.3}\text{Pt}_3\text{O}_4$  were obtained from the smaller unit, *i.e.*, a Picker X-ray unit modified for neutron diffraction. Its monochromatic beam (1.050 Å) is obtained in transmission from the (110) planes of a Be single crystal. Step scans were also obtained in  $0.2^\circ$  steps for a time of approximately 15 min per point. Samples were loaded in cylindrical vanadium cans (5 mil wall thickness) with a diameter of  $3/8$  in. for  $\text{CaPt}_2\text{O}_4$  and 0.25 in. for  $\text{Cd}_{0.3}\text{Pt}_3\text{O}_4$ . The correction for absorption of the data from  $\text{CaPt}_2\text{O}_4$  proved to be negligible; that for  $\text{Cd}_{0.3}\text{Pt}_3\text{O}_4$  was not since the linear absorption coefficient is  $1.25 \text{ cm}^{-1}$ .<sup>11</sup> The usual correction for cylindrical geometry was therefore made, based on the measured packing density of 0.25.

The powder spectra for  $\text{CaPt}_2\text{O}_4$  were fully indexed on the basis of a tetragonal unit cell of dimensions  $a = 5.7786$  (5) and  $c = 5.5989$  (7) Å based on the least-squares refinement of the X-ray line positions calibrated against  $\text{TiCl}_4$  ( $a = 3.8421$  Å).<sup>12</sup> The theoretical density for two formula units in the cell is  $8.8 \text{ g/cm}^3$ . From pycnometric measurements on the extremely finely powdered material a lower limit for the observed density is  $7 \text{ g/cm}^3$ .

The powder spectra for  $\text{Cd}_{0.3}\text{Pt}_3\text{O}_4$  were indexed on the basis of

(1) For the first paper in this series, see D. Cahen, J. A. Ibers, and R. D. Shannon, *Inorg. Chem.*, **11**, 2311 (1972).

(2) (a) Northwestern University; (b) Argonne National Laboratory.

(3) See, for example, Paul N. Rylander, "Catalytic Hydrogenation over Platinum Metals," Academic Press, New York and London, 1967, p 312.

(4) B. Post, private communication; Powder Data File, Card No. 15-520, Joint Committee on Powder Diffraction Standards, Swarthmore, Pa.

(5) C. L. McDaniel, *J. Amer. Ceram. Soc.*, **55**, 426 (1972).

(6) R. Czaya, *Z. Anorg. Allg. Chem.*, **375**, 61 (1970).

(7) (a) J. Waser and E. D. McClanahan, Jr., *J. Chem. Phys.*, **19**, 413 (1951); (b) *ibid.*, **20**, 199 (1952).

(8) D. Cahen, J. A. Ibers, and J. B. Wagner, Jr., submitted to *Inorg. Chem.*

(9) (a) H. Lipson and H. Steeple, "Interpretation of X-Ray Powder Diffraction Patterns," Macmillan, London, 1970, p 94 ff; (b) B. D. Cullity, "Elements of X-Ray Diffraction," Addison-Wesley, Reading, Mass., 1967, p 188 ff.

(10) M. H. Mueller, *Advan. X-Ray Anal.*, **15**, 70 (1972).

(11) K. Weber, *Acta Crystallogr.*, **23**, 720 (1967).

(12) Powder Data File, Card No. 6-0486, Joint Committee on Powder Diffraction Standards, Swarthmore, Pa.

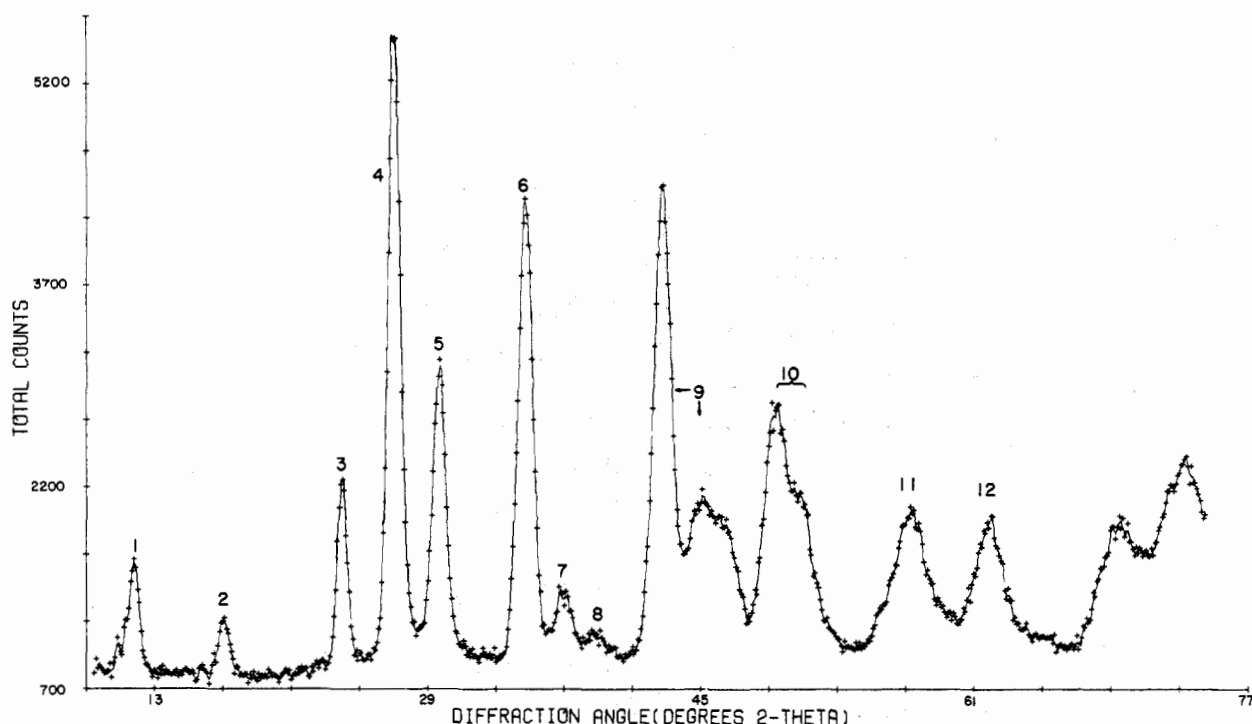


Figure 1. Neutron diffraction data for CaPt<sub>2</sub>O<sub>4</sub>. A five-point smoothed line is drawn through the experimental points. The peak numbers refer to those in Table I.

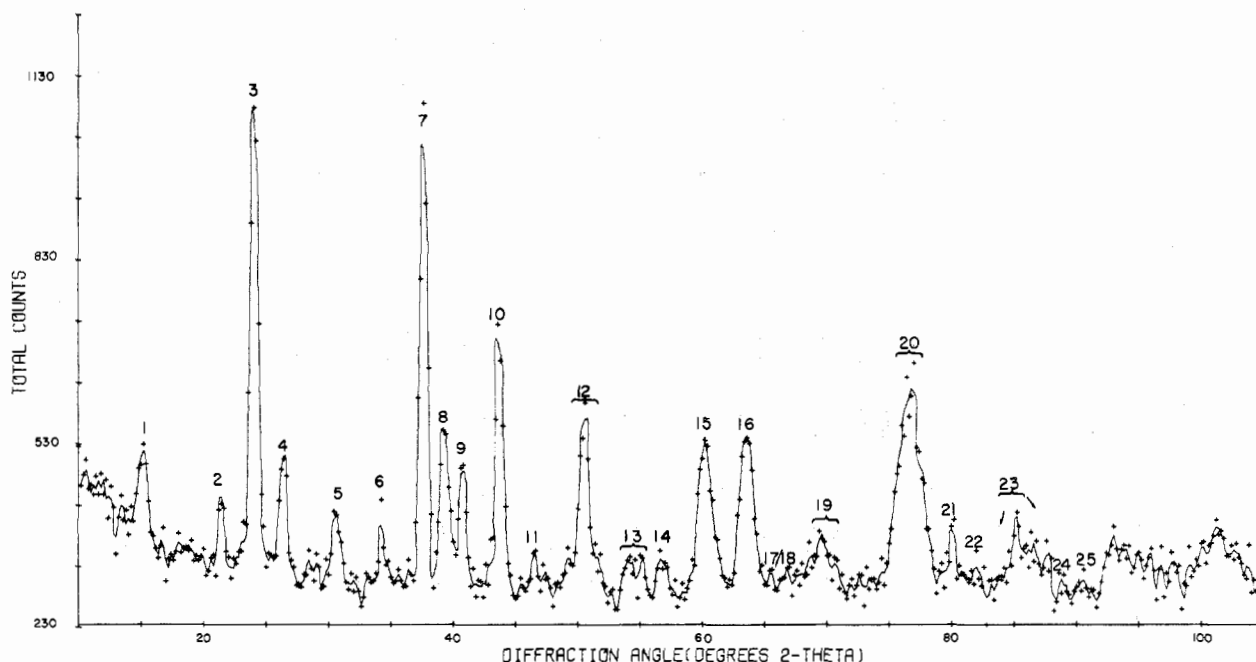


Figure 2. Neutron diffraction data for Cd<sub>0.3</sub>Pt<sub>3</sub>O<sub>4</sub>. A five-point smoothed curve is drawn through the experimental points. The peak numbers refer to those in Table III.

a cubic unit cell with edge 5.6423 (4) Å, again obtained from a least-squares analysis of the calibrated X-ray line positions. This cell edge may be compared with  $a = 5.688$  Å for Cd<sub>1</sub>Pt<sub>3</sub>O<sub>4</sub>. The theoretical density of 12.7 g/cm<sup>3</sup> for four formula units in the cell may be compared with a lower limit to the experimental density of 8 g/cm<sup>3</sup>.

Figures 1 and 2 show the experimental neutron diffraction data for the two compounds. The X-ray data are not shown but are of comparable quality.

Intensities were obtained from the graphed spectra, using a Keuffel and Esser Model 620015 compensating polar planimeter, and standard deviations in these intensities were estimated taking into account the counting statistics and the base levels of the peaks.

Tables I and II give the fully indexed neutron and X-ray powder patterns of CaPt<sub>2</sub>O<sub>4</sub>. The numbers of the observed peaks in Table I refer to those in Figure 1. The  $2\theta$  angle is given in degrees. Thus,

we confirm the tetragonal unit cell proposed by Post.<sup>3</sup> Systematic absences were observed for  $l = 2n + 1$  in the  $hhl$  reflections. The possible space groups are then  $P4_2/mmc$ ,  $P42c$ , or  $P4_2mc$ . Because of the overlap of intensity data we could not distinguish between these possibilities and therefore the space group with highest symmetry,  $D_{4h}^9-P4_2/mmc$ , was chosen.

The powder patterns of Cd<sub>0.3</sub>Pt<sub>3</sub>O<sub>4</sub> for neutrons and X-rays are given in Tables III and IV. The observed intensities in Table III have been corrected for absorption. The pattern fits that of the general Pt bronze structure. The space group is either  $O_h^3-Pm3n$  or  $T_d^4n-P43n$  based on the absence of  $hhl$  for  $l$  odd.

### Structure Refinement

Refinement of the structures was effected through the method of least squares<sup>13</sup> in which the function minimized is  $\Sigma w(I_o - I_c)^2$ , where

TABLE I  
CaF<sub>2</sub> X-Ray Diffraction Data  
 $\lambda = 1.20 \text{ \AA}$

$\#$ (Obs)	2 $\theta$	hkl	$d_c$	$1/d_c^2 \times 10^3$	$E_L \times 10^{-3}$	$I_0 \times 10^{-3}$	$\sigma(I_0) \times 10^{-3}$	$\sigma(I_0)/I_0 \times 10^{-3}$	$d_0$	$d_c$	$1/d_c^2$	$E_L$	$I_0$	$\sigma(I_0)$
1	11.9	100	5.779	106	106	91	1	1.00	5.780	5.779	250	250	113	20
2	17.2	101	4.086	45	154	41	2	1.50	5.899	5.899	0	0	112	10
3	24.0	200	2.889	154	154	107	2	2.25	4.086	4.086	2	2	87	10
4	26.8	210	2.584	31	406	447	6	3.38	4.021	4.021	110	75	93	3
5	27.6	102	2.519	112	253	244	3	3.63	2.367	2.367	280	410	428	7
6	34.2	220	2.045	13	332	374	6	5.06	2.145	2.145	172	173	152	5
7	36.3	300	1.926	50	152	78	8	6.25	2.091	2.091	1	41	29	4
8	38.3	310	1.827	12	30	36	6	8.26	2.042	2.042	41	41	29	4
9	45.6	222	1.650	406	749	819	9	12.25	1.827	1.827	25	27	20	5
10	46.2	312	1.530	7	42.6	42.6	2	16.51	1.776	1.776	15	15	4	2
11	49.1	400	1.445	204	44.4	44.4	0	20.25	1.650	1.650	89	89	38	4
12	50.7	410	1.402	11	50.7	50.7	0	22.20	1.604	1.604	89	89	38	4
13	50.8	004	1.400	106	45.8	45.8	0	22.37	1.597	1.597	10	10	6	4
14	51.1	322	1.391	4	45.8	45.8	0	22.64	1.568	1.568	68	68	86	10
15	52.3	330	1.362	1.4	360	360	7	24.84	1.541	1.541	32	32	36	7
16	52.4	104	1.360	10	52.4	52.4	0	25.25	1.514	1.514	44	44	50	4
17	52.4	411	1.360	11	46.7	46.7	0	25.50	1.445	1.445	38	38	49	3
18	53.2	303	1.340	11	49.1	49.1	0	26.01	1.402	1.402	12	12	16	7
19	53.9	114	1.324	0.6	50.7	50.7	0	26.53	1.398	1.398	18	18	22	7
20	54.7	312	1.306	1.4	55.7	55.7	0	27.06	1.391	1.391	27	27	36	7
21	55.3	420	1.292	55	56.9	56.9	0	27.60	1.360	1.360	12	12	16	7
22	56.9	204	1.260	65	57.2	57.2	0	28.14	1.360	1.360	14	14	18	4
23	57.2	412	1.253	46	58.4	58.4	0	28.68	1.284	1.284	34	34	46	4
24	58.4	214	1.231	12	58.7	58.7	0	29.22	1.259	1.259	72	72	96	2
25	58.7	332	1.225	11	61.5	61.5	0	29.77	1.225	1.225	20	20	28	2
26	61.5	422	1.173	153	62.6	62.6	0	30.42	1.231	1.231	15	15	20	2
27	62.6	300	1.156	0	62.6	62.6	0	31.07	1.225	1.225	2	2	3	12
28	62.6	430	1.156	29	63.4	63.4	0	31.72	1.173	1.173	2	2	1	4
29	63.4	403	1.142	0	63.9	63.9	0	32.37	1.156	1.156	5	5	7	14
30	64.0	304	1.132	31	64.0	64.0	0	33.02	1.156	1.156	6	6	8	14
31	64.0	501	1.132	2	64.7	64.7	0	33.67	1.142	1.142	1	1	0.5	15
32	64.7	413	1.121	9	65.4	65.4	0	34.32	1.133	1.133	1	1	0.5	15
33	65.4	314	1.111	5	66.4	66.4	0	34.97	1.133	1.133	6	6	8	13
34	66.4	511	1.111	0.9				35.62	1.132	1.132	16	16	21	11
35	66.4	511	1.111	0.9				36.27	1.132	1.132	16	16	21	11

TABLE II  
CaF<sub>2</sub> X-Ray Diffraction Data  
 $\lambda = 1.7901 \text{ \AA}$

$\#$ (Obs)	2 $\theta$	hkl	$d_c$	$1/d_c^2$	$E_L$	$I_0$	$\sigma(I_0)$	$\sigma(I_0)/I_0$	$d_0$	$d_c$	$1/d_c^2$	$E_L$	$I_0$	$\sigma(I_0)$
1	5.20	100	5.779	30	250	250	113	20	5.780	5.779	250	250	113	20
2	10.40	200	2.889	12	112	112	87	10	4.086	4.086	2	2	87	10
3	15.60	300	1.926	27	41	41	6	4	1.926	1.926	14	14	18	4
4	18.48	400	1.445	49	30	30	3	3	1.445	1.445	5	5	6	3
5	21.36	500	1.260	65	20	20	2	2	1.260	1.260	8	8	10	2
6	24.24	600	1.167	76	15	15	1	1	1.167	1.167	8	8	10	2
7	27.12	700	1.077	89	11	11	1	1	1.077	1.077	8	8	10	2
8	30.00	800	1.000	100	7	7	1	1	1.000	1.000	10	10	13	5
9	32.88	900	0.933	110	5	5	1	1	0.933	0.933	12	12	16	7
10	35.76	1000	0.870	120	4	4	1	1	0.870	0.870	14	14	18	4

TABLE III  
Ca<sub>2</sub>F<sub>2</sub>O<sub>7</sub> Neutron Diffraction Data  
 $\lambda = 1.050 \text{ \AA}$

$\#$ (Obs)	2 $\theta$	hkl	$d_c$	$1/d_c^2$	$E_L$	$I_0$	$\sigma(I_0)$	$\sigma(I_0)/I_0$	$d_0$	$d_c$	$1/d_c^2$	$E_L$	$I_0$	$\sigma(I_0)$
1	15.1	110	3.990	649	390	390	54	14	3.990	3.990	379	379	390	54
2	21.4	200	2.523	1555	192	192	29	15	2.523	2.523	1555	1555	162	29
3	28.4	210	2.303	1870	234	234	36	20	2.303	2.303	1870	1870	192	29
4	36.4	211	2.303	1870	276	276	47	27	2.303	2.303	1870	1870	234	36
5	44.4	220	2.237	2012	324	324	59	36	2.237	2.237	2012	2012	276	47
6	52.4	220	2.237	2012	396	396	72	44	2.237	2.237	2012	2012	324	59
7	60.4	220	2.237	2012	480	480	89	55	2.237	2.237	2012	2012	396	72
8	68.4	220	2.237	2012	576	576	107	66	2.237	2.237	2012	2012	480	89
9	76.4	220	2.237	2012	684	684	126	78	2.237	2.237	2012	2012	576	107
10	84.4	220	2.237	2012	810	810	147	93	2.237	2.237	2012	2012	684	126
11	92.4	220	2.237	2012	954	954	170	111	2.237	2.237	2012	2012	810	147
12	100.4	220	2.237	2012	1116	1116	195	135	2.237	2.237	2012	2012	954	170
13	108.4	220	2.237	2012	1302	1302	222	180	2.237	2.237	2012	2012	1116	195
14	116.4	220	2.237	2012	1512	1512	261	225	2.237	2.237	2012	2012	1302	222
15	124.4	220	2.237	2012	1746	1746	306	270	2.237	2.237	2012	2012	1512	261
16	132.4	220	2.237	2012	2012	2012	354	324	2.237	2.237	2012	2012	1746	306
17	140.4	220	2.237	2012	2340	2340	405	360	2.237	2.237	2012	2012	2012	354
18	148.4	220	2.237	2012	2754	2754	477	420	2.237	2.237	2012	2012	2340	405
19	156.4	220	2.237	2012	3264	3264	561	504	2.237	2.237	2012	2012	2754	477
20	164.4	220	2.237	2012	3876	3876	657	584	2.237	2.237	2012	2012	3264	561
21	172.4	220	2.237	2012	4500	4500	756	684	2.237	2.237	2012	2012	3876	657
22	180.4	220	2.237	2012	5142	5142	858	792	2.237	2.237	2012	2012	4500	756
23	188.4	220	2.237	2012	5802	5802	975	888	2.237	2.237	2012	2012	5142	858
24	196.4	220	2.237	2012	6480	6480	1107	1008	2.237	2.237	2012	2012	5802	975
25	204.4	220	2.237	2012	7176	7176	1245	1116	2.237	2.237	2012	2012	6480	1107
26	212.4	220	2.237	2012	7890	7890	1398	1248	2.237	2.237	2012	2012	7176	1245
27	220.4	220	2.237	2012	8622	8622	1566	1392	2.237	2.237	2012	2012	7890	1398
28	228.4	220	2.237	2012	9372	9372	1749	1548	2.237	2.237	2012	2012	8622	1566
29	236.4	220	2.237	2012	10140	10140	1937	1716	2.237	2.237	2012	2012	9372	1749
30	244.4	220	2.237	2012	10926	10926	2130	1896	2.237	2.237	2012	2012	10140	1937
31	252.4	220	2.237	2012	11730	11730	2328	2064	2.237	2.237	2012	2012	10926	2130
32	260.4	220	2.237	2012	12552	12552	2531	2232	2.237	2.237	2012	2012	11730	2328
33	268.4	220	2.237	2012	13394	13394	2739	2400	2.237	2.237	2012	2012	12552	2531
34	276.4	220	2.237	2012	14256	14256	2952	2568	2.237	2.237	2012	2012	13394	2739
35	284.4	220	2.237	2012	15138	15138	3171	2736	2.237	2.237	2012	2012	14256	2952
36	292.4	220	2.237	2012	16040	16040	3396	2916	2.237	2.237	2012	2012	15138	2952
37	300.4	220	2.237	2012	16962	16962	3627	3100	2.237	2.237	2012	2012	16040	3396
38	308.4	220	2.237	2012	17904	17904	3864	3288	2.237	2.237	2012	2012	16962	3627
39	316.4	220	2.237	2012	18866	18866	4107	3480	2.237	2.237	2012	2012	17904	3864
40	324.4	220	2.237	2012	19848	19848	4356	3684	2.237	2.237	2012	2012	18866	4107
41	332.4	220	2.237	2012	20850	20850	4611	3894	2.237	2.237	2012	2012	19848	4356
42	340.4													

$I_o$  is the absorption-corrected intensity and where  $I_c$  is the intensity calculated for that particular reflection or group of overlapping reflections and where  $w = 1/\sigma^2(I_o)$ . Calculated intensities were based on the usual Lorentz-polarization factors, multiplicities, and appropriate scattering factors. For the X-ray data the atomic scattering factors for neutral Ca, Cd, Pt, and O atoms were from the calculations of Cromer and Waber.<sup>14</sup> Anomalous dispersion terms for Co K $\alpha$  radiation were taken from Cooper<sup>15</sup> while those for Cu K $\alpha$  radiation were taken from Cromer and Liberman.<sup>16</sup> The neutron scattering factors were from the tabulation by Bacon.<sup>17</sup> Agreement indices are defined as

$$R = (\sum |I_o - I_c| / \sum I_o) \times 100$$

and

$$R_w = ((\sum w(I_o - I_c)^2 / \sum w I_o^2)^{1/2}) \times 100$$

**CaPt<sub>2</sub>O<sub>4</sub>.** Post concluded that the Pt atoms occupy the 4j ( $x, 0, 0$ ) positions in  $P4_2/mmc$ . Our crystallographic data exclude any positions which impose further systematic absences, besides the general condition  $hhl, l = 2n + 1$  (see Table II). This leaves two possible unique positions for the platinum. Crystal packing and approximate bond distance calculations supported Post's initial choice. The neutron diffraction data show that the most likely positions for the oxygen atoms are in  $8n (x, x, 1/4)$ . With Pt and O in these positions, the only chemically reasonable position for the Ca atom is in  $2c (0, 1/2, 0)$ . With the atoms placed in these positions, only two positional parameters need to be determined, namely those of Pt and O. A model showed that the most satisfying structure was obtained if both these parameters were chosen as  $1/4$ . These were the initial parameters for the least-squares refinement. The strategy of refinement took into account the relative scattering powers of the various atoms for X-rays and neutrons (Table V). Thus, the Pt positional parameter (and later the temperature factor as well) was varied in the refinement of the X-ray data. These parameters were then held constant in the refinement of the neutron data, while the oxygen positional parameter (and later its temperature factor) was varied. Initial temperature factors were zero, and only after the positional parameters had been refined were the temperature factors added as variables. Finally an attempt was made to vary the Ca temperature factor, using both data sets separately.

Because of the highly unusual structure that resulted, a structure that contains large voids and somewhat unusual coordination of the oxygen atoms, a difference Fourier synthesis was calculated. Such a synthesis, calculated from overlapping powder diffraction data, will be impossible unless one either omits all overlapping reflections or estimates the intensities for individual reflections which overlap one another. We took the latter approach using the following calculation. If  $I_o$  contains  $h_1, h_2, \dots$ , etc., then

$$I_o(h_1) = I_c(h_1) \frac{\sum_i I_o(h_i)}{\sum_i I_c(h_i)}$$

Here  $\sum_i I_o(h_i)$  is the observed intensity of all the overlapping reflections, and  $\sum_i I_c(h_i)$  is the intensity calculated according to the model structure. From  $I_o(h_1)$ ,  $F_o(h_1)$  is readily derived. The values of  $F_o(h_1)$  along with the calculated phase for  $h_1$  were used in the difference Fourier summation. Because of the smaller number of overlapping reflections this calculation was based on X-ray data. The resulting difference map is featureless as it shows no peaks above  $1 e/\text{\AA}^3$ . We also attempted to vary the Pt occupancy factor and simultaneously to insert some electron density in the position that would correspond to that of a third Pt atom as found in the general bronze structure ( $2e$  or  $4g$ ). No change in the Pt occupancy factor was ob-

(13) A local program, NUPOWLS, was used for the least-squares refinement of the intensity data. This program is based on an earlier version by W. C. Hamilton, as described in W. C. Hamilton and J. A. Ibers, *Acta Crystallogr.*, **16**, 1209 (1963), except for the fact that a diagonal weight matrix (no correlation) was used. In addition to other local programs, the following programs were used: Zalkin's FORDAP, Busing and Levy's ORFFE, and Johnson's ORTEPII.

(14) D. T. Cromer and J. T. Waber in "International Tables for X-Ray Diffraction," Vol. IV, Kynoch Press, Birmingham, England, in press.

(15) M. J. Cooper, *Acta Crystallogr.*, **16**, 1067 (1963).

(16) D. T. Cromer and D. Liberman, *J. Chem. Phys.*, **53**, 1891 (1970).

(17) (a) G. E. Bacon, "Neutron Diffraction," Oxford University Press, London, 1962, pp 31 ff. (b)  $(Cd) = (0.38 + i 0.13)10^{-12}$  cm was used and Cd was treated as an anomalous scatter.

Table V. Relative Contribution to  $F^2$  in Neutron and X-Ray Spectra ( $2\theta = 0^\circ$ )

	CaPt <sub>2</sub> O <sub>4</sub>			Cd <sub>0.3</sub> Pt <sub>3</sub> O <sub>4</sub>		
	% Ca	% Pt	% O	% Cd	% Pt	% O
X-Ray	3	95	2	4	95	1
Neutron	7	53	39	1	66	33

served, and the occupancy factor of an additional Pt atom refined essentially to zero.

The final agreement indices reflect the serious overlap problem encountered and are  $R_w = 16.9$  and  $18.3\%$ , and  $R = 24.6$  and  $16.0\%$  for the X-ray and neutron data, respectively. The final parameters are given in Table VI.

**Cd<sub>0.3</sub>Pt<sub>3</sub>O<sub>4</sub>.** The powder patterns of Cd<sub>0.3</sub>Pt<sub>3</sub>O<sub>4</sub> clearly showed that it is isotypical with Ni<sub>0.25</sub>Pt<sub>3</sub>O<sub>4</sub>.<sup>1</sup> A careful scan was made in order to try to detect any extra peaks that might result from a possible ordering of the Cd ions. No peaks were observed that could not be indexed on the basis of the simple Pt bronze structure.

We refined the X-ray data using the positional parameters of the centrosymmetric Ni bronze structure. Subsequently, the neutron data were refined both in space groups  $Pm3n$  and in the noncentrosymmetric group  $P43n$ , the only difference being the positional parameters of the oxygen atoms [ $(1/4, 1/4, 1/4)$  in  $Pm3n$  and  $(x, x, x)$  in  $P43n$ ]. The value of  $R_w$  obtained for  $x = 0.24$  was  $13.3\%$  and was higher for  $x$  values farther away from  $0.25$ , while it was  $11.4\%$  for  $x = 0.25$ . Thus the structure seems to be centrosymmetric. (A noncentrosymmetric bronze-like structure has been reported for Ag<sub>3</sub>PO<sub>4</sub> and Ag<sub>3</sub>AsO<sub>4</sub>.<sup>18</sup>)

Because of the nonstoichiometry of Cd<sub>0.3</sub>Pt<sub>3</sub>O<sub>4</sub> the Cd occupancy factor was varied in the refinement of the X-ray data set. The final value ( $0.31 \pm 0.05$ ) agrees reasonably well with that obtained from wet chemical analysis ( $0.22 \pm 0.02$ ). In the refinement of the X-ray data, the oxygen temperature factor was not varied, while with the neutron data the Cd occupancy factor was held constant. The final agreement indices,  $R_w$  and  $R$ , were  $11.4$  and  $9.8\%$  for the neutron data and  $8.7$  and  $6.8\%$  for the X-ray data. The final parameters obtained from these refinements are given in Table VII.

## Discussion

Figure 3 shows an overall view of the crystal structure of CaPt<sub>2</sub>O<sub>4</sub>. The result is a highly-unusual arrangement and, as far as we know, a unique one. The striking features of the structure are the nonintersecting Pt chains in the  $x$  and  $y$  direction, the loose packing, and the coordination of the oxygen atoms. Many two-dimensional chain compounds are known, but all of these are better described as layer compounds [e.g., TaS<sub>2</sub>, HgPbP<sub>14</sub>, and silicates such as Mg<sub>3</sub>(Si<sub>4</sub>O<sub>10</sub>)(OH)<sub>2</sub>].<sup>19</sup> The only other known material with an arrangement resembling that of CaPt<sub>2</sub>O<sub>4</sub> is Hg<sub>2.85</sub>AsF<sub>6</sub>, which has Hg chains in two dimensions.<sup>20</sup> Pt chain compounds in one and three dimensions are known<sup>1,21</sup> and CaPt<sub>2</sub>O<sub>4</sub> forms the missing two-dimensional link.

The AB<sub>2</sub>O<sub>4</sub> stoichiometry is a quite common one for ternary oxides. Kugimiya and Steinfink<sup>22</sup> have reviewed the crystal chemistry of these oxides. They propose a scheme for grouping the AB<sub>2</sub>O<sub>4</sub> oxides into several known structure types, based on a plot of a "force constant" vs. radius ratio. Their force constant is defined as

$$K_{ab} = X_a X_b / r_e^2$$

where  $r_e^2 = (r_a + r_o)^2 + (r_b + r_o)^2 + 1.55(r_a + r_o)(r_b + r_o)$  and the radius ratio is  $r_a/r_b$ .  $X_a$  and  $X_b$  are the electronegativities of A and B. For CaPt<sub>2</sub>O<sub>4</sub> one obtains a value of  $0.91$  for  $K_{ab}$  and  $1.62$  for  $r_a/r_b$ .<sup>23</sup> This places the com-

(18) R. W. G. Wyckoff, "Crystal Structures," Vol. 3, 2nd ed, Interscience, New York, N. Y., 1965, p 152.

(19) See, for example, H. Krebs, "Fundamentals of Inorganic Crystal Chemistry," P. H. L. Walter, translator, McGraw-Hill, London, 1968.

(20) R. J. Gillespie, Abstracts, 165th National Meeting of the American Chemical Society, Dallas, Texas, April 1973, No. INOR-1.

(21) K. Krogmann, *Angew. Chem., Int. Ed. Engl.*, **8**, 35 (1969).

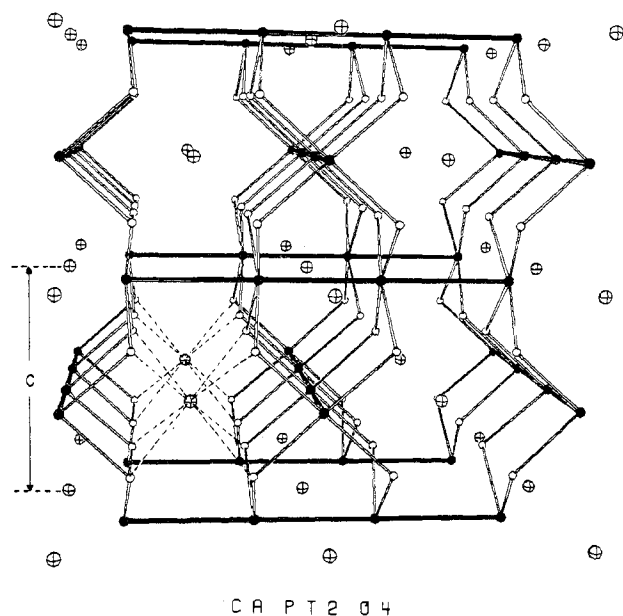
(22) K. Kugimiya and H. Steinfink, *Inorg. Chem.*, **7**, 1762 (1968).

**Table VI.** Final Parameters for  $\text{CaPt}_2\text{O}_4$  from Neutron and X-Ray Data

	x	y	z	Wyckoff notation	Site symmetry	B, Å <sup>2</sup>
Ca	0	1/2	0	2c	mmm	3.9 (2.2) <sup>a</sup>
Pt	0.242 (4)	0	0	4j	mm	0.7 (2)
O	0.268 (7)	0.268	1/4	8n	2	0.7 (9)

<sup>a</sup> Errors in parentheses.**Table VII.**  $\text{Cd}_{0.3}\text{Pt}_3\text{O}_4$  Final Parameters from X-Ray and Neutron Data

	x	y	z	Occupancy	B, Å <sup>2</sup>	Wyckoff notation	Site symmetry
Cd	0	0	0	0.31 (3) <sup>a</sup>	2.5 <sup>b</sup>	2a	m3
Pt	1/4	0	1/2	3 <sup>b</sup>	0.6 (2)	6c	42m
O	1/4	1/4	1/4	4 <sup>b</sup>	1.9 (1)	8e	32

<sup>a</sup> Errors in parentheses. <sup>b</sup> Not varied in refinement.**Figure 3.** Crystal structure of  $\text{CaPt}_2\text{O}_4$ . The platinum and oxygen atoms are drawn as solid and open circles, respectively. The coordination of oxygen atoms around Ca is indicated in the lower left corner.

pound on the borderline between the  $\text{BaAl}_2\text{O}_4$  and normal spinel structure in their diagram. Although this might indicate some unusual behavior for  $\text{CaPt}_2\text{O}_4$ , it does not explain its structure which bears no resemblance to either of the two adjacent structure types.

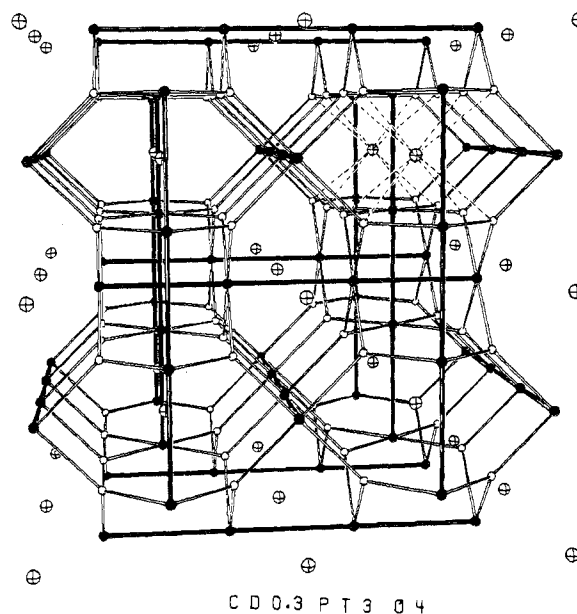
$\text{CaRh}_2\text{O}_4$  has been reported to possess the  $\text{CaFe}_2\text{O}_4$  structure.<sup>24</sup> Muller-Buschbaum and von Schenck have argued that  $\text{A}^{\text{IV}}\text{B}^{\text{III}}_2\text{O}_4$  compounds will tend to attain the  $\text{CaFe}_2\text{O}_4$  structure if  $r_a > r_b$ , a condition clearly fulfilled for  $\text{CaPt}_2\text{O}_4$ .<sup>25</sup>

It seems clear that the structure of  $\text{CaPt}_2\text{O}_4$  is not readily explained if considered separately from those of other platinum bronzes. If one compares Figure 4, which shows the structure of  $\text{Cd}_{0.3}\text{Pt}_3\text{O}_4$ , the general bronze structure, with

(23) All ionic radii in this discussion were taken or adapted from the compilations by R. D. Shannon and C. T. Prewitt, *Acta Crystallogr. Sect. B*, **25**, 925 (1969); **26**, 1046 (1970).  $r[\text{Ca}^{2+}(\text{VIII})] = 1.12 \text{ \AA}$ ;  $r[\text{Cd}^{2+}(\text{VIII})] = 1.07 \text{ \AA}$ ;  $r[\text{O}^{2-}] = 1.40 \text{ \AA}$ ;  $r[\text{Pt}^{2+}(\text{IV})] = 0.60 \text{ \AA}$ ;  $r[\text{Pt}^{3+}(\text{IV})] = 0.69 \text{ \AA}$ . The Roman numerals indicate the coordination number of the ion.

(24) J. Pruedhomme, *C. R. Acad. Sci., Ser. C*, **271**, 1073 (1970).

(25) Hk. Muller-Buschbaum and R. von Schenck, *Z. Anorg. Allg. Chem.*, **377**, 70 (1970).

**Figure 4.** Crystal structure of  $\text{Cd}_{0.3}\text{Pt}_3\text{O}_4$ . The oxygen atoms are drawn as open circles and the platinum atoms as solid ones. The coordination of oxygen atoms around Cd is indicated in the upper right corner.**Table VIII.** Selected Bond Distances (Å) and Angles (deg) in  $\text{CaPt}_2\text{O}_4$ 

		Distances			
Pt <sub>2</sub> -Pt <sub>2</sub>	2.79 (5) <sup>a</sup>	1 <sup>b</sup>	O-O	2.68 (8)	2
Pt <sub>1</sub> -Pt <sub>3</sub>	2.99 (5)	1		2.7994 (3)	2
Pt-O	2.09 (3)	4		3.09 (8)	2
Pt-Ca <sub>1</sub>	3.17 (1)	2	Ca-O	2.481 (3)	8
Pt-Ca <sub>2</sub>	3.21 (1)	2			
		Angles			
O-Pt-O	95 (1)	2	Pt <sub>1</sub> -O-Pt <sub>2</sub>	110 (3)	1
	171 (2)	2	Ca-O-Ca	173 (3)	1
	84 (2)	2	O-Ca-O	65 (2)	4
O-Pt <sub>1</sub> -Pt <sub>2</sub>	94 (1)	4		68.7 (1)	4
O-Pt <sub>1</sub> -Pt <sub>3</sub>	86 (1)	4		77 (2)	4
Pt-O-Ca	87.4 (9)	2		102 (2)	4
	88.7 (9)	2		113 (1)	4
				114 (2)	4
				180 (0)	4

<sup>a</sup> Errors in parentheses. <sup>b</sup> Number of equivalent distances (or angles).**Table IX.** Interatomic Distances (Å) in  $\text{Cd}_{0.3}\text{Pt}_3\text{O}_4$ 

Pt-O	1.9948 (2) <sup>a</sup>	4 <sup>b</sup>	O-O	2.8211 (2)	6
Pt-Pt	2.8211 (2)	2	O-Cd	2.4432 (2)	8
Pt-Cd	3.1541 (2)	4			

<sup>a</sup> Errors in parentheses. <sup>b</sup> Number of equivalent distances.

Figure 3, one sees a striking similarity. By removing the chains of Pt atoms in the z direction from  $\text{Cd}_{0.3}\text{Pt}_3\text{O}_4$ , we essentially derive the  $\text{CaPt}_2\text{O}_4$  structure. The reason for the unusual behavior of  $\text{Ca}^{2+}$  is not clear as " $\text{CaPt}_3\text{O}_4$ " would fit very well in the bronze stability field. One factor that might cause  $\text{CaPt}_2\text{O}_4$  to behave differently is its relatively high formation temperature and its exceptional stability as compared with that of other Pt bronzes.<sup>8</sup>

**Bonding in  $\text{CaPt}_2\text{O}_4$  and  $\text{Cd}_{0.3}\text{Pt}_3\text{O}_4$ .** Tables VIII and IX summarize the bond distances and angles for  $\text{CaPt}_2\text{O}_4$  and  $\text{Cd}_{0.3}\text{Pt}_3\text{O}_4$ . No unusual distances or coordination are seen in  $\text{Cd}_{0.3}\text{Pt}_3\text{O}_4$ , except that here, contrary to the Ni ions in  $\text{Ni}_{0.25}\text{Pt}_3\text{O}_4$ , the Cd ions do fit very well in the channels formed by the oxygen atoms. The sum of the  $\text{Cd}^{2+}$  (VIII) and  $\text{O}^{2-}$  ionic radii is 2.47 Å, which compares well with the

observed distance of 2.44 Å. Similarly, the Pt-O distance of 1.99 Å compares well with  $\text{Pt}^{2+}(\text{IV}) + (\text{O}^{2-}) = 2.00 \text{ \AA}$ . The  $\text{sp}^2$  hybridization of the oxygen atoms accounts readily for the planar-triangular bonding to Pt. The Pt-Pt distance indicates metallic bonding (Pt-Pt in Pt metal is 2.80 Å).

In  $\text{CaPt}_2\text{O}_4$  the situation seems to be more complex. The Pt atoms are not equidistant but slightly paired, although we presume that all the bonds have metallic character. The  $\text{Ca}^{2+}$  ions again fit very well in the oxygen channels, as shown by the Ca-O bond length of 2.48 Å (the sum of the ionic radii is 2.52 Å). The oxygen coordination around the Pt is slightly distorted from square planarity, but hardly significantly so. The Pt lies 0.15 (5) Å from the plane defined by the four-coordinated oxygen atoms. As in the general bronze structure, chains of alternating Ca and O atoms run diagonally through the crystal.

Most unusual is the coordination of Pt around oxygen. It was this feature that prompted us to calculate the difference Fourier synthesis and to vary the Pt occupancy factor (as described above) in order to see if some electron density might be found in the channels of the structure. The resulting bonding shows each oxygen atom connected to two platinum atoms, with the angle between them close to  $120^\circ$  [110

(3)<sup>o</sup>]. This suggests that some electron density centered on the oxygen is pointing toward the position where Pt would be found in the bronze structure. Although such bonding for oxygen is not uncommon (e.g.,  $\text{H}_2\text{O}$ ) it is highly unusual in a metal oxide. As mentioned before, metallic bonding between the platinum atoms is found in both  $\text{Cd}_{0.3}\text{Pt}_3\text{O}_4$  and  $\text{CaPt}_2\text{O}_4$ . We are currently investigating some of the physical properties of these and other bronzes in order to look for evidence of metallic behavior in these materials. The results of this study together with details on the preparation and catalytic behavior of the bronzes will be the subjects of future publications.

Registry No.  $\text{CaPt}_2\text{O}_4$ , 39394-82-6;  $\text{CdPt}_3\text{O}_4$ , 39394-83-7;  $\text{Cd}_x\text{-Pt}_3\text{O}_4$ , 39394-68-8.

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Contribution from the Department of Chemistry and Geology, Clemson University, Clemson, South Carolina 29631

## Solid Coordination Compounds of Alkali Metal Salts with a Neutral Aromatic Amine.<sup>1</sup> I. Molecular Structure and Bonding

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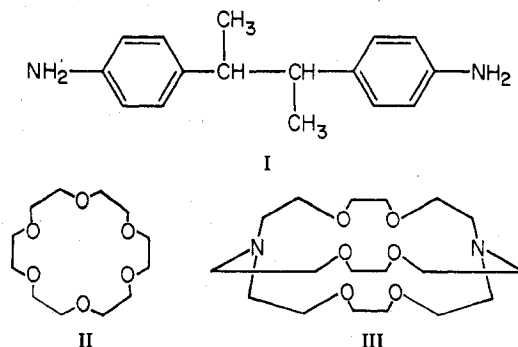
Uni-univalent metal salts, including alkali metal salts, are precipitated from aqueous solution by complexation with racemic *p,p'*-diamino-2,3-diphenylbutane. The stoichiometry and infrared and solution properties of these compounds are presented and discussed in terms of coordination compounds in which the metal ion is octahedrally coordinated by six amino nitrogen atoms to give an infinite three-dimensional polymeric structure. Evidence is presented for strong metal ion-amino nitrogen bonding as well as interaction between the anion and amino hydrogens. Both of these attractive interactions are attributed to ion dipole forces. Evidence is also presented which strongly indicates that there is little anion-cation interaction.

### Introduction

In a previous communication,<sup>1</sup> we reported initial results on the precipitation of simple alkali metal salts such as sodium chloride from aqueous solution by complexation with racemic *p,p'*-diamino-2,3-diphenylbutane (I). The stoichiometry, constant composition, and infrared spectrum of the NaCl adduct with I was interpreted in terms of a structure in which six nitrogen atoms from six amino groups are octahedrally coordinated about the sodium. This is unusual since of the alkali metals only lithium salts are known to form isolable coordination compounds with nitrogen donor ligands. The more stable are those where the ligand is ammonia, and replacement of ammonia with amines decreases their stability. It is therefore surprising that I, a weakly basic and bulky aromatic amine, can exhibit bonding with sodium.

Unlike nitrogen, there are many examples of oxygen donor ligands which effectively coordinate with alkali metal cations. Some notable recent examples are the macrocyclic ethers

(crown ethers) (II)<sup>2</sup> and the macroheterobicyclics (cryptates) (III).<sup>3</sup> Subsequent to our initial report methylenedianiline (IV), a closely analogous compound of I, was reported to form isolable compounds with alkali metal salts.<sup>4</sup>



(1) (a) A preliminary account of a portion of this work has been reported: N. P. Marullo and R. A. Lloyd, *J. Amer. Chem. Soc.*, **88**, 1076 (1966). (b) Abstracted in part from the Master's thesis of R. A. Lloyd and Ph.D. thesis of G. T. Cochran, Clemson University, 1967.

(2) D. J. Sam and H. E. Simmons, *J. Amer. Chem. Soc.*, **94**, 4024 (1972), and references cited therein.

(3) J. M. Lehn and J. P. Sauvage, *Chem. Commun.*, 440 (1971), and references cited therein.

(4) T. C. Shields, *Chem. Commun.*, 832 (1968).